

# Organic electroluminescent devices: enhanced carrier injection using SAM derivatized ITO electrodes†

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Taking as a device model ITO|TPD|Alq<sub>3</sub>|Al (where TPD is *N,N*-bis(3-methylphenyl)-*N,N'*-diphenyl-1,1'-biphenyl-4,4'-diamine and Alq<sub>3</sub> is tris(quinolin-8-olato)aluminium) it is shown that control and improvement of carrier injection may be achieved using self-assembled monolayers (SAMs) to manipulate the Schottky energy barrier at the ITO–TPD interface. By using polar adsorbate molecules with the dipole oriented outward from the surface an artificial dipolar layer is formed and the work function is increased, and *vice versa*. With this method the threshold voltage for light emission (turn-on) can be reduced by 4 V and the maximum luminance increased by a factor of 3.5, giving an overall performance superior to that using the more stable Ag/Mg counter electrode. The SAMs effect is confirmed using a Scanning Kelvin Probe (SKP) to profile the relative work function of half-coated ITO samples. Increases in work function in excess of 0.3 eV are observed, in line with predictions using the calculated molecular dipoles of the SAM molecules.

Organic electroluminescent devices (OEDs) based on either thin films of polymers or lower molecular weight materials show increasing promise as commercial devices for use in full colour, flat panel displays. However, some problems still remain, particularly the high operating voltage, low efficiency and short device lifetime.<sup>1</sup>

The performance of the OEDs is strongly influenced by the conditions at both electrodes, under which carrier injection takes place. The rate of injection into a bilayer device depends on the position of the work function of the electrode relative to the appropriate band of the adjacent organic layer, the field distribution within the device and the quality of the contact. The electron injecting electrode requires a low work function metal such as magnesium or calcium. However, these materials tend to be readily oxidised, a factor which limits device lifetime. Aluminium is more stable but has a higher work function. Conversely, for hole injection, a high work function is required. Since it is generally necessary for this electrode to be optically transparent, indium tin oxide (ITO) coated glass is almost universally used. Although its work function has been found to be as high as 5.15 eV it appears that there may still be a barrier to hole injection.

In response, much work has been published on methods to improve the injecting properties of the electrodes. In the case of aluminium, it has been reported that a thin (5–24 Å thick) insulating buffer layer, such as LiF<sup>2</sup> or Al<sub>2</sub>O<sub>3</sub>,<sup>3</sup> between the aluminium electrode and the emitter layer can dramatically enhance the electron injection and the efficiency of OEDs. The improvement has been attributed to the removal of trapping states at the interface and better alignment of the metallic Fermi level with the conduction band of the emitter. Multilayer Langmuir–Blodgett films have also been used with similar effect.<sup>4,5</sup> Work with ITO has concentrated on the use of plasma<sup>6–8</sup> and aqua regia<sup>8,9</sup> treatments to improve carrier injection. Increased performance is believed to be due to the elimination of surface contaminants, improved contact with the hole transport layer because of changes in surface morphology, and a change in the work function.

It is known that the work function of a metal is influenced by

the electrostatic conditions at its surface.<sup>10</sup> Using this effect, Zehner<sup>11,12</sup> and Campbell<sup>13,14</sup> have recently shown that dipolar arene thiols form self-assembled monolayers on gold and can be exploited to control the work function. This suggests that SAM modified electrodes could greatly enhance the performance of OEDs. Taking as a model device ITO|TPD|Alq<sub>3</sub>|Al, SAMs can be incorporated at the ITO–TPD interface. It has been demonstrated that both carboxylic acid, and preferably phosphonic acid (PA) derivatives, form SAMs on ITO.<sup>15</sup> Nüesch *et al.*<sup>16</sup> observed improved performance in single layer polymer devices using a carboxylic acid functionalised oligomer monolayer grafted to the ITO electrode. They interpreted this effect as being due to reduced pinhole formation at the interface as well as an increase in the ITO work function.

Using a similar approach we demonstrated that pre-treatment of the ITO electrode with a dipolar phosphonic acid, (2-chloroethyl)phosphonic acid, greatly enhanced the performance of the device, most notably by reduction of the turn-on. Preliminary results were published in a conference report.<sup>17</sup> This paper contains a more detailed study covering a wider range of compounds and includes measurement of the effect on the work function using a Scanning Kelvin Probe (SKP). Most significantly, (4-nitrophenyl)phosphonic acid increases the work function by more than 0.3 eV and reduces the turn-on voltage by 4 V giving a device performance superior to that with the Mg/Ag counter electrode.

## Experimental

### Substrate preparation

The substrate (precoated ITO glass) was purchased from Balzers (Baltracom 247 ITO, 1.1 mm thick) with a sheet resistance and transmittance of 20 Ω sq<sup>-1</sup> and 90% respectively. This was cut into 20 × 25 mm sample slides.

**Etching.** For device fabrication it was necessary to etch a strip of ITO 6 mm wide from the slide (Fig. 1) to prevent electrical shorting in the final device. A protective layer of commercial (positive) photoresist was applied to the area to be preserved and the exposed ITO was then etched in a solution of

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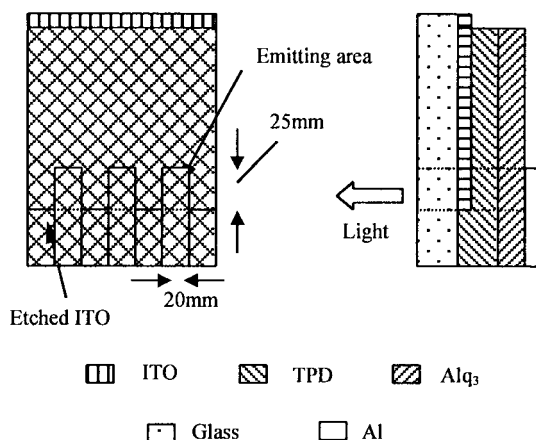


Fig. 1 Basic three-cell, bilayer device structure used in the study.

1 : 1 hydrochloric acid and distilled water. Following etching, the resist layer was removed with acetone.

**Cleaning.** The slides were thoroughly cleaned by swabbing with acetone followed by a sequence of 30-minute ultrasonic baths in detergent (Decon 90), de-ionised water, 1,1,1-trichloroethane, and finally acetone. The cleaned slides were sonicated and then stored under ethanol until required. Immediately prior to use, the slides were further cleaned by suspension in hot acetone vapour.

**SAM treatment.** Four adsorbate phosphonic acids (PAs) (Fig. 2) were selected by dipole moment and purchased from SALOR and Aldrich. The dipole moment of each PA was calculated *via* a semi-empirical Hartree-Fock geometry optimisation and single point calculation at the PM3 level of approximation. For comparative purposes the PA group was substituted for a methyl group because of the inability of the PM3 approximation to cope adequately with P-O bonds.

Solutions of each PA ( $1 \times 10^{-4}$  M) were prepared in 30% methanol in chloroform (30MCF), with the exception of AMPA for which an aqueous solution was used. For device fabrication, the slides were derivatized with a SAM by immersion in a solution of the appropriate PA, typically for a period of 16 hours. For the SKP measurements only half of the slide was treated. After derivatization, the slides were removed from solution and rinsed thoroughly with fresh 30MCF. The presence of each PA on the ITO surface was confirmed by measuring the advancing contact angle relative to bare ITO of triply distilled water on the surface (Table 1). Our previous work has verified the adsorption of similar PAs on ITO using Secondary Ion Mass Spectroscopy (SIMS)<sup>18</sup> and Electron Paramagnetic Resonance (EPR).<sup>19</sup>

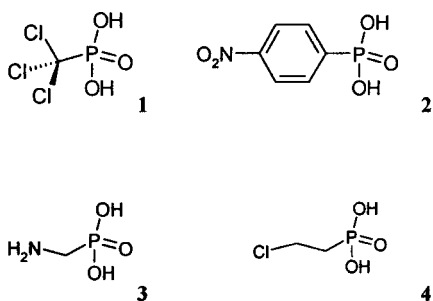


Fig. 2 Molecular structures of the four SAM adsorbates used in the study. 1. (trichloromethyl)phosphonic acid (TCPA), 2. (4-nitrophenyl)phosphonic acid (4-NPPA), 3. (aminomethyl)phosphonic acid (AMPA), and 4. (2-chloroethyl)phosphonic acid (2-CEPA).

Table 1 Triply distilled water advancing contact angles on derivatized ITO substrates

Substrate	Contact angle/ $^{\circ}$ using triply distilled water
ITO (as received)	$73 \pm 2$
ITO(2-chloroethyl)phosphonic acid	$49 \pm 2$
ITO(aminomethyl)phosphonic acid	$55 \pm 2$
ITO(trichloromethyl)phosphonic acid	$64 \pm 2$
ITO(4-nitrophenyl)phosphonic acid	$67 \pm 3$

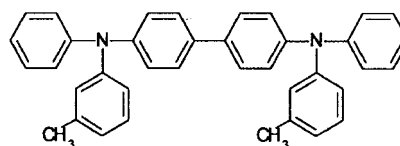


Fig. 3 *N,N'*-Bis(3-methylphenyl)-*N,N'*-diphenyl-1,1'-biphenyl-4,4'-diamine (TPD).

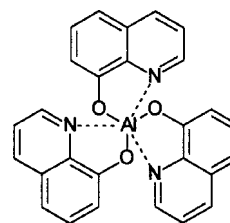


Fig. 4 Tris(quinolin-8-olato)aluminium ( $\text{Alq}_3$ ).

## Devices

**Device fabrication and characterisation.** The OEDs were fabricated by successive vapour deposition of the organic layers onto derivatized ITO glass followed by three patterned aluminium electrodes using an Edwards E306 vacuum coating unit. The organic constituents of the bilayer device consisted of a hole transport layer of *N,N'*-bis(3-methylphenyl)-*N,N'*-diphenyl-1,1'-biphenyl-4,4'-diamine (TPD) (Fig. 3) purchased from Aldrich, and an electron transport and emitter layer of tris(quinolin-8-olato)aluminium ( $\text{Alq}_3$ ) (Fig. 4). The  $\text{Alq}_3$  was synthesised in the laboratory using established methods<sup>20,21</sup> and purified by three-cycle entrainer sublimation.<sup>22</sup> The basic device design (Fig. 1) consists of three cells each with an emitting area of  $\sim 5 \text{ mm}^2$  that could be tested independently. The substrates, four treated and one untreated with each PA, were used to fabricate five devices. Each organic layer (500 Å thick) was vacuum deposited at a rate of  $2\text{--}4 \text{ \AA s}^{-1}$  and a base pressure of  $1 \times 10^{-6}$  mbar. Finally three adjacent top electrodes of aluminium were vacuum evaporated onto the organic layers at a rate of  $5\text{--}10 \text{ \AA s}^{-1}$  and patterned using a shadow mask. For comparative purposes a single device was fabricated using a Mg/Ag alloy (ratio 10 : 1) as the top electrode.

Following fabrication the devices were transferred immediately from the vacuum chamber for characterisation. The current-voltage ( $I$ - $V$ ) and luminance-voltage ( $L$ - $V$ ) characteristics were recorded under forward bias using a Keithley 195 electrometer, a Thurlby PL320 power source and a Macam Q102 radiometer interfaced to a personal computer. The characteristics of the three cells were then averaged.

## Work function measurements

**Scanning Kelvin probe (SKP) measurements.** Work function changes were measured using a piezo-ceramic based vibrating capacitor technique (or Kelvin probe<sup>23,24</sup>) with a gold mesh reference electrode (Besocke Delta Phi model 'S')<sup>25</sup> mounted in a purpose built vacuum chamber. Briefly, the Kelvin probe method comprises a reference electrode positioned a few millimetres away from the sample surface which together

represent the plates of a capacitor. The two are then connected through a high resistance  $R$  and there is a net transfer of charge as the two Fermi levels come into coincidence. The potential difference developed across the two electrodes is equal to the difference in work functions. By vibrating the tip of the reference electrode the capacitance of the system is changed and there is a flow of charge from one plate to the other through  $R$ , detected as a voltage drop across it. A nulling potential  $V$  can then be applied to the electrodes such that no voltage drop is recorded across  $R$ . At this point the potential applied is equal and opposite to the difference in work function of the two metals.

The sample was mounted on a micromanipulator arm within a vacuum system such that the surface could be scanned across the face of the reference electrode using a stepper motor controlled by a personal computer. The entire SKP assembly, controller software and electronics were designed and built in the School of Chemistry at the University of Nottingham. Typical performance of the SKP showed noise levels of  $<1$  mV ( $\sim 0.3\%$  of the values measured) and a drift of  $<5$  mV  $\text{h}^{-1}$  at a resonant frequency of 237 Hz using a vibration amplitude of  $\sim 1.0$  mm when operating under medium vacuum ( $2 \times 10^{-6}$  mbar) conditions. The scan sequence consisted of five averaged probe readings taken at 100 points along the 25 mm length of the slide. The scans were repeated three times and averaged to produce a final differential work function profile for the slide. By scanning a slide partially treated with a SAM, the difference in work function between treated and untreated ITO is determined directly, obviating the need for a separate reference surface of known work function.

## Results and discussion

### Scanning Kelvin Probe studies

Fig. 5 shows the SKP profiles for both the bare and partially treated substrates. The effect of the monolayers is seen as a shift in the relative contact potential approximately midway across the substrate. In an idealised situation we would expect to see a sharp step function at the transition between the treated and untreated regions of the substrate. However, the limited linear resolution of the probe together with surface concentration effects in the derivatisation solution mean that this step is significantly broadened.

It was first necessary to determine the immersion time required for complete monolayer formation and hence to reproduce the maximum effect on the work function. A study of a series of slides treated over a 24-hour period is shown in Fig. 6 for the case of (2-chloroethyl)phosphonic acid. 16 hours was therefore adopted as the derivatisation time.

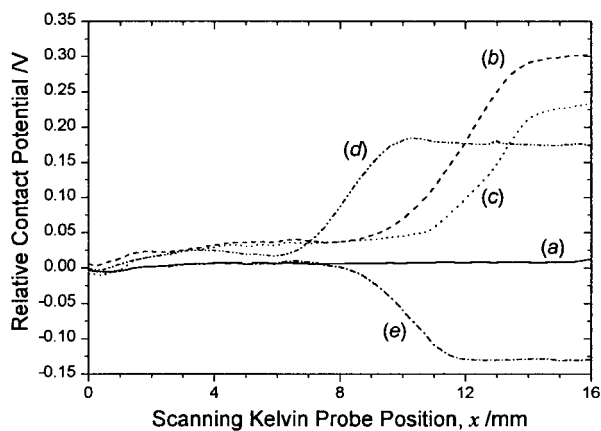


Fig. 5 Relative work function profiles for each partially treated substrate: (a) no SAM, (b) 4-NPPA, (c) TCPA, (d) 2-CEPA and (e) AMPA.

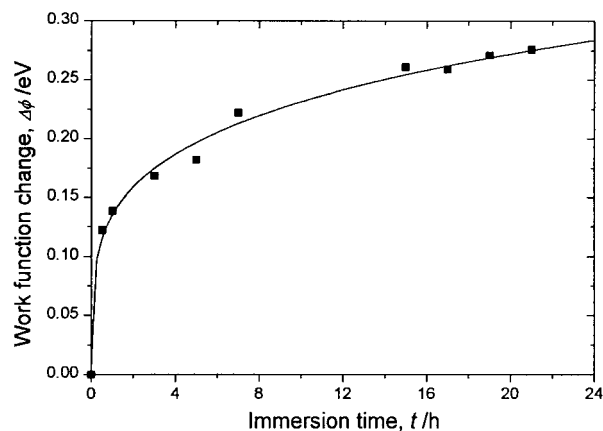


Fig. 6 Dependence of  $\Delta\phi$  on immersion time for 2-(chloroethyl)phosphonic acid.

Following immersion it is desirable to wash the slide in excess pure solvent to prevent recrystallisation of the PA on the surface. This is a process that takes only a few seconds. Measurements after extended washing periods show that the PA molecules slowly desorb, confirming the necessity for rapid washing. This may be of importance in connection with electrode patterning. If a photoresist mask were to be used, for example, the solvent for its removal would need to be one in which the PA was insoluble.

The work function shift  $\Delta\phi$  induced by a uniform dipolar surface layer is determined by the change in electrostatic potential ( $\Delta V$ ) created at the surface (Fig. 7) and can be derived from classical electrostatics as eqn. (1),

$$\Delta\phi = -e\Delta V = -eN \left( \frac{\mu_{\text{mol}\perp}}{\epsilon_r \epsilon_0} \right) \quad (1)$$

where  $N$  is the surface number density (taken to be  $1.7 \times 10^{-14}$   $\text{cm}^{-2}$  as determined by Wrighton<sup>15</sup>),  $\epsilon_r$  is the dielectric constant of the SAM (taken to be 2.5<sup>13,14,26</sup>), and  $\mu_{\text{mol}\perp}$  is the molecular dipole normal to the surface (defined as

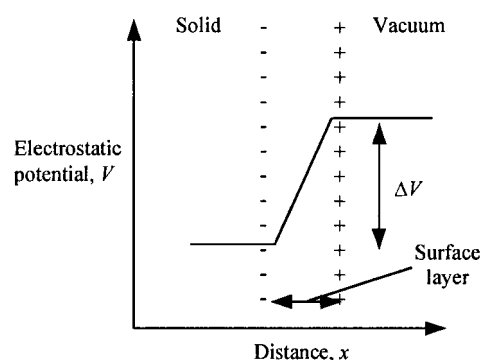


Fig. 7 Variation in electrostatic potential in a surface dipolar layer. Here the dipole is directed into the surface reducing the work function.  $\Delta V$  can be calculated using eqn. (1).

Table 2 Calculated and measured work functions. The dipole moments were calculated using a semi-empirical Hartree-Fock geometry optimisation and single point calculation at the PM3 level of approximation

SAM	Work function change, $\Delta\phi$ / eV	Calculated dipole, $\mu_{\text{mol}\perp}$ / D	Calculated work function change, $\Delta\phi$ / eV
4-NPPA	0.303	5.73	0.720
TCPA	0.185	1.76	0.221
2-CEPA	0.179	1.69	0.212
AMPA	-0.140	-1.43	-0.179

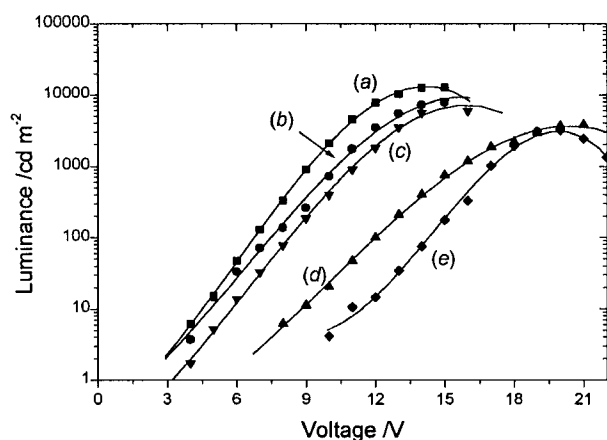
positive when pointing inwards). Table 2 shows the average empirical work function shift  $\Delta\phi$  and the calculated molecular dipoles with the corresponding predicted work function shift.

Using this simple model, the changes in work function are in the same sequence and direction as those predicted by the calculations, despite the approximations involved. In particular, when the direction of the dipole is reversed so too is the direction of change of the work function, suggesting that the effect is, at least to a first approximation, as described by the model in eqn. (1). More accurate calculations require knowledge of the molecular conformation, coverage, the molecular order of the monolayer, and the nature of the bonding to the surface, as well as an accurate value of the dipole moment in the adsorbed state.

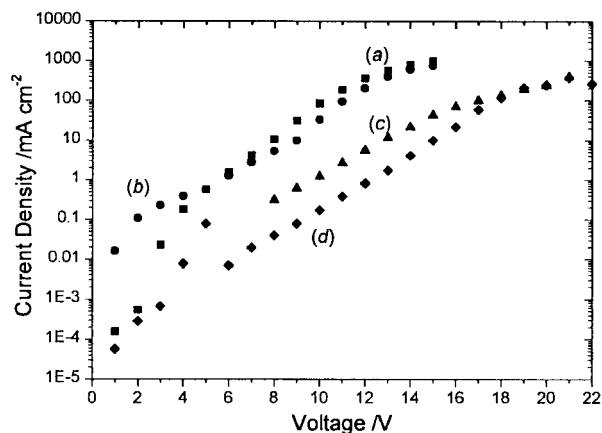
### Device studies

The  $L-V$  and  $I-V$  plots are given in Fig. 8 and 9 respectively. The most significant characteristics are summarised in Table 3.

The  $I-V$  plots show that the SAMs have a significant effect on the magnitude of the injected current. In the  $L-V$  plots for the 4-NPPA, TCPA and 2-CEPA SAMs there is a dramatic improvement in the luminance threshold voltage over the basic device, to the extent that the performance is superior to devices using Mg/Ag as the counter electrode.<sup>17</sup> This suggests that there is a considerable effect on the Schottky barrier attributable to the dipole of the surface molecules. This is further confirmed by the adverse effect of the AMPA SAM. Since the turn-on voltage is governed by the onset of two-carrier injection, this also indicates that, in the aluminium device, injection of holes from the ITO anode occurs secondary to injection of electrons from the cathode. It is noted, however, that while the 4-NPPA device does exhibit improved luminance and efficiency, there is no significant decrease of the luminance threshold over the TCPA device despite significantly different dipole moments. This may imply that the conditions have been met whereby the work function of the ITO is greater than that of the TPD such that the contact becomes ohmic. As such, a further increase in work function would not necessarily result in



**Fig. 8** Luminance–voltage characteristics for devices using: (a) 4-NPPA, (b) TCPA, (c) 2-CEPA, (d) no SAM, and (e) AMPA.



**Fig. 9** Current–voltage characteristics for devices using: (a) 4-NPPA, (b) TCPA, (c) no SAM, and (d) AMPA.

a lower threshold. The work function of ITO has been reported to be 4.5–5.15 eV, however, accurate values of the TPD Fermi level are to the best of our knowledge not available. However the energy of the HOMO has been measured to be around 5.4 eV. This suggests a barrier height of approximately 0.25 eV when bare ITO is used. Alternatively, this may indicate that below some critical barrier height, the ITO becomes the primary injection electrode and the turn-on voltage is now determined by the Al electrode. Consequently there is no net reduction in threshold with further increase of the ITO work function. Nevertheless, assuming that electrons are the dominant carrier, there will still be an accompanying increase in hole injection leading to higher recombination efficiency and light output.

The introduction of a monolayer at the ITO interface will also have consequences for the internal field distribution, which may be beneficial for hole injection. Indeed, it becomes necessary to invoke an explanation that considers the field distribution within the device in order to explain the effect of SAMs on the Mg/Ag device. Our previous work<sup>17</sup> has shown that SAMs on ITO produce only a minor reduction in threshold when Mg/Ag is the counter electrode. Our proposed explanation is based on a dynamic voltage dependent internal field distribution. If the device is viewed as a set of resistances in series representing the interfaces, at low voltages the potential difference is predominantly across the more resistive electrode. As the voltage is increased the more resistive electrode becomes injecting and the potential difference starts to increase across the other electrode until this becomes injecting. This is the threshold for two carrier injection and hence light emission. Thus, the threshold corresponds to the voltage for the onset of injection at the less resistive electrode. When Mg/Ag is used, the more resistive electrode at low voltages is assumed to be the ITO. The use of a SAM on the ITO improves its hole injection, and reduces its resistance, but not below that of the Mg/Ag. Consequently, the effect is small. The situation is very different when aluminium is the electron injecting electrode. The higher work function and the presence of an oxide layer make this the more resistive electrode at low voltages. The turn-on voltage is now determined by the ITO. Enhancement of its hole injecting

**Table 3** Characteristic device performance info using derivatized ITO electrodes. Device structure: ITO|SAM|TPD|Alq<sub>3</sub>|Al

Device	Threshold voltage/V	Max. luminance/cd m <sup>-2</sup>	Drive voltage @ 300 cd m <sup>-2</sup> /V	Q.E. @ 300 cd m <sup>-2</sup> (%)	L.E. @ 300 cd m <sup>-2</sup> /lm W <sup>-1</sup>
4-NPPA	2.0	13 120@14 V	7.88	0.91	1.28
TCPA	2.0	9037@15.6 V	8.82	0.69	0.81
2-CEPA	4.0	7120@15.7 V	9.51	0.63	0.92
No SAM	6.0	3640@20.5 V	13.60	0.61	0.56
AMPA	9.5	3010@20 V	15.50	0.46	0.30

properties by the use of a SAM therefore has a more dramatic effect.

## Conclusion

Self-assembled monolayers of dipolar phosphonic acids adsorbed onto the surface of ITO glass provide a cheap and simple way of modifying the work function. Measurements using a Scanning Kelvin Probe show that the changes are of magnitude and direction broadly as predicted from theory. Electroluminescent devices using the modified electrodes show greatly enhanced performance particularly in threshold voltage.

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